## Preparation of Uniform Hexapod Cu<sub>2</sub>O and Hollow Hexapod CuO

## Young-Sik Cho and Young-Duk Huh\*

Department of Integrated Molecular Sciences Chemistry, Institute of Nanosensor and Biotechnology, Dankook University, Gyeonggi-Do 448-701, Korea. \*E-mail: ydhuh@dankook.ac.kr Received May 30, 2013, Accepted July 1, 2013

Key Words: Hexapod, Cu<sub>2</sub>O, Hollow hexapod, CuO, Thermal oxidation

The atomic arrangement of the outer surface of inorganic oxides is strongly dependent on the morphology of inorganic oxides. The morphology-dependent properties of inorganic materials, such as magnetic, photocatalytic, and antibacterial activities, is one of the most important experimental issues in inorganic technology. <sup>1-5</sup> Many researchers have prepared uniform and specific shaped inorganic oxides to understand the morphology-dependent properties of inorganic oxides.

Most morphology-controlled synthesis has focused on the one-dimensional inorganic oxides, such as wires, rods, and ribbons with the assistance of hard-templates and soft-templates. Three-dimensional inorganic oxides with closed shapes such as cubes, octahedrons, cuboctahedrons, and rhombic dodecahedrons are usually prepared by hydrothermal and solvothermal reactions. However, few investigations have been carried out on three-dimensional inorganic oxides with open shapes such as tetrapods, hexapods, octapods, and star-like shapes. 15-19

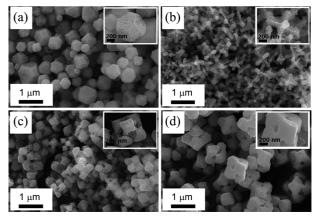
In general, the hollow metal oxides have superior catalytic efficiencies to non-hollow metal oxides, due to its low density, high surface-to-volume ratio, and large void spaces. 20,21 The hollow CuO were widely used as catalysts, anodes for lithium batteries, and gas sensors.<sup>22-24</sup> The hollow CuO spheres were fabricated using a droplet-to particle aerosol spray pyrolysis method.<sup>25</sup> The hollow CuO spheres were synthesized using a hydrothermal reaction of carbohydrated copper precursors.<sup>26</sup> Hollow CuO microspheres were synthesized via a complex assisted method with hexamethylenetetramine as a complex agent.<sup>27</sup> Hierarchically porous CuO hollow spheres were also synthesized via a simple one-pot template-free method.<sup>24</sup> CuO nanotubes were prepared by simple thermal oxidation of copper nanowires.<sup>28</sup> However, most of the hollow CuO had microsphere structures. It is very difficult to synthesize a uniform morphology of the three-dimensional hollowed structures of CuO. To the best of our knowledge, this hollowed hexapod CuO is the first CuO synthesized until now.

Cu<sub>2</sub>O is one of the most important inorganic oxides that have been extensively investigated for morphology-controlled synthesis. <sup>29-32</sup> Cu<sub>2</sub>O is considered to be the best candidate for the preparation of unique morphology. We also investigated the morphology evolution of Cu<sub>2</sub>O from the thinhexapod to the aggregated sphere *via* a thick-hexapod, truncated hexapod, truncated octahedron, and cuboctahedron. <sup>33</sup> However, the shape of the hexapod was not perfect and the

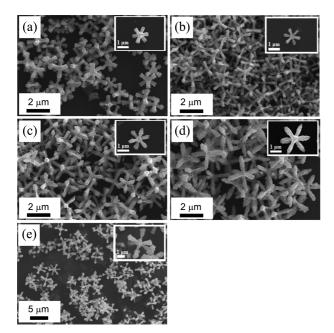
end of each pod was not straight. Therefore, we reinvestigated the synthetic process for the preparation of uniform hexapod-like Cu<sub>2</sub>O. We also present a simple thermal oxidation method for the hollow hexapod of CuO from the direct oxidation of hexapod Cu<sub>2</sub>O in air at a high temperature.

For the preparation of uniform hexapod Cu<sub>2</sub>O, we revisited our earlier work of the morphology evolution for Cu<sub>2</sub>O.<sup>33</sup> Especially, we reexamined the synthetic conditions for the hexapod Cu<sub>2</sub>O. Even though the same chemical composition was used for the preparation of Cu<sub>2</sub>O, we used a different type of microwave oven from that used in the earlier work. The morphology of Cu<sub>2</sub>O is strongly dependent on the irradiation time of the microwave. The irradiation time of the microwave was adjusted for the preparation of uniform hexapod Cu<sub>2</sub>O. Figure 1 shows the SEM images of the four Cu<sub>2</sub>O products using the simple microwave-assisted reaction with different irradiation times with other conditions kept the same. As the irradiation time increases, the morphologies of Cu<sub>2</sub>O evolve from the truncated octahedron to the octahedron concaved at the center of the surfaces, via an almost hexapod shape and a hexapod with thin pyramid arms at the end of each pod. Therefore, the microwave irradiation time of 120 s was adequate for the synthesis of the uniform hexapod morphology of Cu<sub>2</sub>O.

Since most microwave-assisted reactions can be completed within a few minutes, the microwave-assisted method is an attractive, simple and convenient synthetic technique. The final temperature of the reaction solution was increased

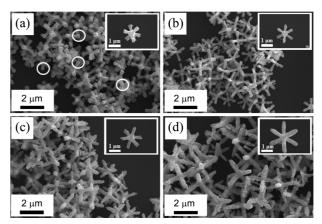


**Figure 1.** SEM images of the Cu<sub>2</sub>O products prepared using the microwave-assisted method with various irradiation times; (a) 100 s, (b) 120 s, (c) 140 s, and (d) 160 s.



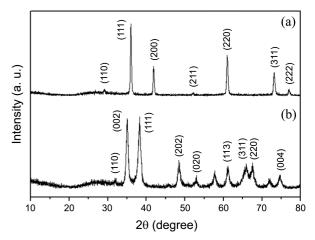
**Figure 2.** SEM images of the  $Cu_2O$  products prepared under the irradiation of microwave for 120 s. The cooling rates were adjusted by adding different amounts of cold water to the 150 mL of  $Cu_2O$  product solutions; (a) 200 mL, (b) 100 mL, (c) 50 mL, and (d) no cold water. (e) SEM images of thin-hexapod  $Cu_2O$  prepared in our earlier work.<sup>33</sup>

to 90 °C within 120 s under microwave irradiation, the Cu<sub>2</sub>O crystal grew in a short period. However, the thermal diffusion of the reaction medium may affect the fine morphology of the Cu<sub>2</sub>O crystals. We attempted to control the thermal quenching rates of thermal diffusion by quickly pouring various amounts of cold water into the final solutions after microwave irradiation. Figure 2 shows the SEM images of the four Cu<sub>2</sub>O products using different amounts of cold water with other conditions kept the same. All four products show similar morphologies of hexapod Cu<sub>2</sub>O. Therefore, the hexapod structures of Cu<sub>2</sub>O crystals were formed within a few minutes by thermal heating of the microwave irradiation. However, the shapes of the four products were slightly changed and the length of each pod was gradually elongated on a small scale as the amount of cold water decreased. The end of the each pod was formed by aggregating the small particles, as shown in Figure 2(d). This indicates that minor changes of the morphologies were formed by the thermal diffusion of the reaction medium after microwave irradiation. With a decrease in the amounts of cold water added to the final products, the thermal diffusion rate increased. The reaction process at the end of the pods occurred to increase the length of each pod, while keeping the shape of the hexapod unchanged, as shown in Figure 2. For comparison, the thin-hexapod Cu<sub>2</sub>O prepared in our earlier work is shown in Figure 2(e).<sup>33</sup> We concluded that the experimental conditions of Figure 2(a) were the optimal conditions for the preparation of the uniform and perfect hexapod of Cu<sub>2</sub>O. The average length and diameter of each pod is 800 nm and 300 nm, respectively.



**Figure 3.** SEM images of the CuO products prepared from the direct thermal oxidation in the air at 400 °C for 5 h with the Cu<sub>2</sub>O precursors, the cooling rates of which were adjusted by adding different amounts of cold water to the 150 mL of Cu<sub>2</sub>O product solutions; (a) 200 mL, (b) 100 mL, (c) 50 mL, and (d) no cold water.

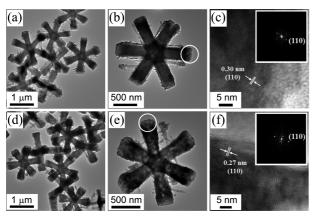
Figure 3 shows the SEM images of the four CuO products prepared by the direct thermal oxidation of the four Cu<sub>2</sub>O precursors in air at 400 °C for 5 h. The outer shapes of the CuO products resemble those of Cu<sub>2</sub>O precursors. However, the surfaces of the hexapod were rather roughened. Moreover, the ends of each pod showed opened structures as shown in the circle area in Figure 3(a). This indicates that CuO products have the hollowed structures. The hollow hexapod of CuO products obtained by the thermal oxidation of hexapod Cu<sub>2</sub>O precursors was confirmed by the XRD measurement. Figure 4 shows the XRD patterns of the hexapod Cu<sub>2</sub>O precursors and the hollow hexapod of the CuO products. The observed hexapod Cu<sub>2</sub>O and hollow hexapod of CuO matched with the reported data of the cubic Cu<sub>2</sub>O



**Figure 4.** XRD patterns and Miller indices of SEM images of (a) hexapod Cu<sub>2</sub>O prepared under the irradiation of microwave for 120 s and 200 mL cold water added to the 150 mL of product solution and (b) hollow hexapod CuO prepared from the direct thermal oxidation in the air at 400 °C for 5 h with the Cu<sub>2</sub>O precursors prepared under the irradiation of microwave for 120 s and 200 mL cold water added to the 150 mL of the product solution.

(JCPDS 05-0667, a = 0.4269 nm) and monoclinic CuO (JCPDS 45-0937, a = 0.4685 nm, b = 0.3425 nm, c = 0.5130 nm, and  $\beta = 99.549^{\circ}$ ), respectively. No impurities in the hexapod Cu<sub>2</sub>O and the hollow hexapod of CuO in the XRD patterns indicated that pure Cu<sub>2</sub>O and CuO were formed, respectively. The formation process of the Cu<sub>2</sub>O product is followed by the consecutive reactions of the first step for the formation of Cu(OH)<sub>2</sub> by reaction with Cu<sup>2+</sup> ion and OH ions and then the second step for the dehydration and reduction to form Cu<sub>2</sub>O by using the glucose as an reducing agent. The hollow hexapod of CuO was formed by the direct thermal oxidation of hexapod Cu<sub>2</sub>O precursors reacted with oxygen in the air.

Checking TEM images for the verification of hollowed materials is more effective than checking SEM images. Figure 5 shows the TEM images of hexapod Cu<sub>2</sub>O precursors and the hollow hexapod of CuO products. Most of the arms of the hexapod Cu<sub>2</sub>O show a dark region in the TEM images, indicating that the inside of the arms of the hexapod Cu<sub>2</sub>O is filled, as shown in Figures 5(a) and (b). The lattice spacing of 0.30 nm corresponding to the (110) plane of Cu<sub>2</sub>O was confirmed from the fringe pattern of the hexapod Cu<sub>2</sub>O, as shown in Figure 5(c). The inset in Figure 5(c) showed the fast Fourier transform (FFT) pattern of the hexapod Cu<sub>2</sub>O. However, the inside of the arms of the hexapod CuO show a relatively light region compared to the outside of the arms of the hexapod CuO, as shown in Figures 5(d) and (e). Similarly, the FFT pattern and lattice spacing of 0.27 nm corresponding to the (110) plane of CuO was also observed, as shown in Figure 5(f). Therefore, the hollow hexapod structures of CuO were formed by the thermal oxidation of the filled hexapod structures of Cu<sub>2</sub>O. These hollowed structures may be formed by the diffusion of the inner parts of the Cu<sub>2</sub>O particles in the hexapod to the outer parts of the hexa-



**Figure 5.** (a, b) HRTEM image and (c) high-magnification HRTEM image of hexapod Cu<sub>2</sub>O prepared under the irradiation of microwave for 120 s and 200 mL cold water added to the 150 mL of the Cu<sub>2</sub>O product solution. (d, e) HRTEM images and (f) high-magnification HRTEM image of hollow hexapod CuO prepared from the direct thermal oxidation in the air at 400 °C for 5 h with the Cu<sub>2</sub>O precursor prepared under the irradiation of microwave for 120 s and 200 mL cold water added to the 150 mL of the product solution. The insets of (c) and (f) show the FFT patterns of an individual hexapod Cu<sub>2</sub>O and hollow hexapod CuO, respectively.

pod.  $\text{Cu}_2\text{O}$  particles of the outer parts in the hexapod thermally reacted with oxygen in the air at a high temperature to produce CuO as the oxidation products. This diffusion behavior is known as a type of Kirkendall effect in accordance with Fick's first diffusion law.<sup>34</sup>

In conclusion, the filled hexapod Cu<sub>2</sub>O precursors were also prepared under microwave irradiation for only 120 s using a commercial microwave oven. The optimal experimental conditions for the perfect and uniform hexapod-like Cu<sub>2</sub>O precursors were examined. The control of the cooling rate by adding cold water was also examined for the elimination of further crystal growth at the end of the arm of the hexapod Cu<sub>2</sub>O precursors due to the thermal diffusion in reaction medium. The uniform hollow structure and hexapod CuO products were also prepared from the direct thermal oxidation of the filled hexapod Cu<sub>2</sub>O precursors.

## **Experimental Section**

CuCl<sub>2</sub>·2H<sub>2</sub>O (99%, Aldrich), polyethylene glycol (PEG, Mw 20,000, Fluka), NaOH (97%, Aldrich), and glucose (ACS reagent, Aldrich) were used as received. For the preparation of hexapod Cu<sub>2</sub>O, 50 mL of a 0.01 M glucose aqueous solution and 2.4 g PEG were added to 10 mL of a 0.2 M CuCl<sub>2</sub> aqueous solution. 3 mL of 2.0 M NaOH aqueous solution was added to the mixed solution. 87 mL of deionized water was added to the mixed solution in a 500 mL beaker. A final mixed solution was placed into a commercial microwave oven (Magic MWO-230KD, 2.45 GHz, 800 W). The irradiation time of the microwave was changed from 100 s to 160 s to investigate the morphology changes with time. The final temperature of the mixed solution after microwave irradiation was 90 °C. In order to reduce the thermal diffusion effect in the reaction solution after microwave irradiation, the cold water was quickly added to the reaction solution. Various amounts (200 mL, 100 mL, and 50 mL) of cold water of 10 °C were added to the mixed solution to adjust the quenching rate for the thermal diffusion of the solution. After 30 min, the products were washed several times with ethanol and water. And the products were dried at 60 °C for 12 h in a drying oven. The structure and morphology of the Cu<sub>2</sub>O and CuO products were characterized by powder Xray diffraction (XRD, PANalytical, X'pert-pro MPD), scanning electron microscopy (SEM, Hitachi S-4300), and highresolution transmission electron microscopy (HRTEM, JEOL JEM-3010).

**Acknowledgments.** This study was supported by the research fund of Dankook University in 2013.

## References

- Burda, C.; Chen, X.; Narayanan, R.; El-Sayed, M. A. Chem. Rev. 2005, 105, 1025.
- Jun, Y. W.; Lee, J. H.; Choi, J. S.; Cheon, J. J. Phys. Chem. B 2005, 109, 14795.
- 3. Xu, H.; Wang, W.; Zhu, W. J. Phys. Chem. B 2006, 110, 13829.
- 4. Wang, X.; Wu, H. F.; Kuang, Q.; Huang, R. B.; Xie, Z. X.; Zheng,

- L. S. Langmuir 2010, 26, 2774.
- Hua, Q.; Shang, D.; Zhang, W.; Chen, K.; Chang, S.; Ma, Y.; Jiang, Z.; Yang, J.; Huang, W. *Langmuir* 2011, 27, 665.
- Xia, Y.; Yang, P.; Sun, Y.; Wu, Y.; Mayers, B.; Gates, B.; Yin, Y.; Kim, F.; Yan, H. Adv. Mater. 2003, 15, 353.
- 7. Wang, X.; Li, Y. Inorg. Chem. 2006, 45, 7522.
- Rao, C. N. R.; Deepak, F. L.; Gundiah, G.; Govindaraj, A. Pro. Solid State Chem. 2003, 31, 5.
- Tiano, A. L.; Koenigsmann, C.; Santulli, A. C. Wong, S. S. Chem. Commun. 2010, 8093.
- Wang, Z.; Chen, X.; Liu, J.; Mo, M.; Yang, L.; Qian, Y. Solid State Commun. 2004, 130, 585.
- Kuo, C. H.; Chen, C. H.; Huang, M. H. Adv. Funct. Mater. 2007, 17, 3773.
- 12. Siegfried, M. J.; Choi, K. S. J. Am. Chem. Soc. 2006, 128, 10356.
- 13. Kuo, C. H.; Huang, M. H. J. Phys. Chem. C 2008, 112, 18355.
- Zhao, H. Y.; Wang, Y. F.; Zeng, J. H. Cryst. Growth Des. 2008, 8, 3731
- Chen, Z. Z.; Shi, E. W.; Zheng, Y. Q.; Li, W. J.; Xiao, B.; Zhuang, J. Y. J. Cryst. Growth 2003, 249, 294.
- Kim, M. J.; Cho, Y. S.; Park, S. H.; Huh, Y. D. Cryst. Growth Des. 2012, 12, 4180.
- Zhang, X.; Xie, Y.; Xu, F.; Xu, D.; Liu, H. Can. J. Chem. 2004, 82, 1341.

- 18. Chang, Y.; Zeng, H. C. Cryst. Growth Des. 2004, 4, 273.
- 19. Liang, Z. H.; Zhu, Y. J. Mater. Lett. 2005, 59, 2423.
- 20. Luo, X. W.; Archer, L. A.; Yang, Z. Adv. Mater. 2008, 20, 3987.
- Shao, Q.; Wang, L. Y.; Wang, X. J.; Yang, M. C.; Ge, S. S.; Yang, X. K.; Wang, J. X. Solid State Sci. 2013, 20, 29.
- 22. Ju, J. H.; Ryu, K. S. J. Electrochem. Soc. 2011, 158, A814.
- 23. Zhu, J.; Qian, X. J. Solid State Chem. 2010, 183, 1632.
- 24. Qin, Y.; Zhang, F.; Chen, Y.; Zhou, Y.; Li, J.; Zhu, A.; Luo, Y.; Tian, Y.; Yang, J. *J. Phys. Chem. C* **2012**, *116*, 11994.
- Jian, G.; Liu, L.; Zachariah, M. R. Adv. Funct. Mater. 2013, 23, 1341.
- Titirici, M. M.; Antonietti, M.; Thomas A. Chem. Mater. 2006, 18, 3808.
- Zhang, Y.; Wang, S.; Qian, Y.; Zhang, Z. Solid State Sci. 2006, 8, 462
- 28. Cho, Y. S.; Huh, Y. D. Bull. Korean Chem. Soc. 2008, 29, 2525.
- 29. Kuo, C. H.; Huang, M. H. Nano Today 2010, 5, 106.
- 30. Xu, J.; Xue, D. Acta Mater. 2007, 55, 2397.
- 31. Choi, K. S. Dalton Trans. 2008, 5432.
- 32. Prabhakaran, G.; Murugan, R. CrystEngComm 2012, 14, 8338.
- 33. Lee, Y. J.; Huh, Y. D. Mater. Res. Bull. 2011, 46, 1892.
- Yin, Y.; Erdonmez, C. K.; Cabot, A.; Hughes, S.; Alivisatos, A. P. Adv. Funct. Mater. 2006, 16, 1389.